

Journal of Fluorine Chemistry 112 (2001) 165-171



www.elsevier.com/locate/jfluchem

# Room temperature preparations of second and third transition series tetrafluorides and a possible novel structure type for OsF<sub>4</sub> and RhF<sub>4</sub>

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Received 15 June 2001; accepted 31 August 2001

Dedicated to Dr. Karl O. Christe on the occasion of his 65th birthday

#### Abstract

The polymeric tetrafluorides,  $MF_4$  (M = Mo, Ru, Pd, Re, and Os) have been precipitated from their  $MF_6^{2-}$  salts in liquid anhydrous hydrogen fluoride (aHF) at  $\sim 20$  °C. For M = Ru, Os and Pd,  $AsF_5$  can be used to displace the tetrafluoride, but this strong  $F^-$  acceptor is too strongly oxidizing for use in making  $MoF_4$  and  $ReF_4$ , for which  $SbF_5$  is effective. X-ray powder diffraction photograph (XRDP) show the  $MF_4$  to have the same structures as those from higher temperature preparations, when previously known. XRDP of  $OsF_4$  are very like those of  $RhF_4$  and indicate a close structural relationship of the pair, and a significant structural difference from the  $PdF_4$  type structure adopted by M = Pd, Pt, Ir, and Re. A new, two-dimensional, 'raft' structure is proposed for  $OsF_4$  and  $RhF_4$ . © 2001 Published by Elsevier Science B.V.

Keywords: Syntheses; Tetrafluorides; Crystal structures; RhF4; OsF4; ReF4

# 1. Introduction

The binary fluorides of the second and third transition series elements provide a wide range of oxidation states for each element, especially for the platinum metals. For Ru, Rh, and Ir the known range is from M<sup>VI</sup> to M<sup>III</sup>. Since the coordination number is commonly six, this means that the structures vary from the three-dimensional F-bridged network of the trifluorides [1], to the molecular hexafluorides [2]. This has provided for investigation of the effect of oxidation state (and the d electron configuration) on M–F bonding in these elements [3].

In the tetrafluorides, the two F ligands that are non-bridging can be *trans* as in NbF<sub>4</sub> [4], and RuF<sub>4</sub> [3] or *cis* as in IrF<sub>4</sub> [5], PtF<sub>4</sub>. (Here, the unit cell given was a pseudomonoclinic cell with a = b, this being half the diagonal of the true orthorhombic F cell [6,7], and PdF<sub>4</sub> [8,9]). The M–F–M bridging angle can also vary widely, being nearly linear in NbF<sub>4</sub> [4] and close to 135° in RuF<sub>4</sub> [3] and PdF<sub>4</sub> [8,9]. Indeed such variations, and the absence of a predictively useful explanation for them, prompted efforts to prepare other tetrafluorides, when a convenient new synthetic approach was found [10]. This provided authentic

OsF<sub>4</sub>, and ReF<sub>4</sub>, for the first time and also an easier route to previously known tetrafluorides including MoF<sub>4</sub> [11], RuF<sub>4</sub> [12], RhF<sub>4</sub> [9] and PdF<sub>4</sub> [9].

X-ray powder diffraction photograph (XRDP) of  $OsF_4$  closely resembled those of  $RhF_4$  and were, in important aspects, different from those having the  $PdF_4$  type structure [8,9], i.e. M = Ir, Pt, Pd and Re. This prompted a new evaluation of structural possibilities for  $RhF_4$  and its structural relative  $OsF_4$ . The new preparative method for these tetrafluorides, and a possible structural model for the Rh and  $OsF_4$  are the subjects of this paper.

# 2. Results and discussion

The appropriate second and third transition element  $[MF_6]^{2-}$  salts were prepared as previously described [13], and the tetrafluorides were derived from them by precipitation, at room temperature, from aHF solutions, using various acids. The choice of acid was determined by (1) convenience, (2) the  $F^-$  acceptor strength of the desired tetrafluoride, and (3) the redox limitations.

Because they are gases, BF<sub>3</sub>, and AsF<sub>5</sub>, are both convenient F<sup>-</sup>acceptors. (GeF<sub>4</sub> is another, which proved to be excellent for AgF<sub>3</sub> precipitation [14], but it was not tried in

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this context.) Unfortunately, BF<sub>3</sub> is not a very strong F<sup>-</sup> acceptor [15,16], and for related reasons not very soluble in aHF. It proved to be ineffective in producing the tetrafluorides. With the more easily oxidized tetrafluorides, ReF<sub>4</sub> and MoF<sub>4</sub>, the AsF<sub>5</sub> was too potent an oxidizer and rapidly gave  $M^V$  products; so for these tetrafluorides it was necessary to employ SbF<sub>5</sub> as the F<sup>-</sup> acceptor. The syntheses can therefore be summarized by the following equations:

$$\begin{split} [MF_6]^{2-} + 2AsF_5 \to MF_4 \downarrow + 2[AsF_6]^-, \quad M = Os, Ru \text{ and Pd} \\ & (1) \\ [MF_6]^{2-} + 2SbF_5 \to MF_4 \downarrow + 2[SbF_6]^-, \quad M = Re \text{ and } Mo \end{split}$$

Because the  $[MF_6]^{2-}$  salts used in this work were largely potassium salts, deriving from the earlier study [13], these syntheses involved much tedious washing with aHF in which the KAsF<sub>6</sub> and KSbF<sub>6</sub> are of low solubility. For any future work it would be worthwhile to prepare the Li<sub>2</sub>MF<sub>6</sub> salts since LiAsF<sub>6</sub> and LiSbF<sub>6</sub> are much more soluble in aHF than their potassium counterparts.

As already noted ReF<sub>4</sub> and MoF<sub>4</sub> reduce AsF<sub>5</sub>, indicating that the  $M^{\rm IV}$  oxidation state at the left hand end of each transition series is at least a moderately reducing one. But with the nuclear charge increase along the series we quickly pass to tetrafluorides that are stable with respect to AsF<sub>5</sub>. In PdF<sub>4</sub> we have a tetrafluoride capable of oxidizing XeF<sub>2</sub> to XeF<sub>4</sub> in aHF at room temperature [17].

Even RhF<sub>4</sub> is easily reduced. It is pertinent that as Sharpe pointed out long ago,  $[RhF_6]^{2-}$  salts [18], like those of  $[PdF_6]^{2-}$  [19], are easily made in solution in BrF<sub>3</sub> made basic with alkali fluoride. But in neutral BrF<sub>3</sub> a pink complex is formed ([18] and present work), which is probably BrF<sub>3</sub>·RhF<sub>3</sub>, since it is diamagnetic, and its pyrolysis gives the purple–red pyrochlore-form of RhF<sub>3</sub>. This purple–red product had been identified in Sharpe's early work [18] as RhF<sub>4</sub>. The pyrochlore-form of RhF<sub>3</sub> is akin to its FeF<sub>3</sub> relative reported by dePape and Ferey [20], and to (rhombohedral) pyrochlore-form NiF<sub>3</sub> which is generated [21] in the slow thermolysis of NiF<sub>4</sub> above -60 °C.

XRDP of the tetrafluorides made by this low temperature route showed that the same structures pertained as in those same fluorides made by other methods. These photographs disclosed the variety of structure types set out in Fig. 1.

All of the patterns were indexed except that of MoF<sub>4</sub> the strong-line pattern of which, however did agree with the d

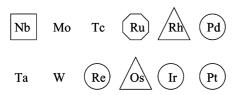


Fig. 1. Observed structure types of the second and third transition series tetrafluorides. SnF<sub>4</sub> type ( $\square$ ), [4]; RuF<sub>4</sub> type ( $\bigcirc$ ), [3]; PdF<sub>4</sub> type ( $\bigcirc$ ), [8,9]; possible OsF<sub>4</sub> type ( $\triangle$ ), [9] and this work.

spacings given by Payne and Asprey [11]. Curiously, OsF<sub>4</sub> showed an almost identical XRDP to that of RhF<sub>4</sub> and distinctly different from that of its pseudo-isoelectronic relative RuF<sub>4</sub>. It had been noted in the early work [9] that the XRDP of RhF<sub>4</sub> exhibited puzzling differences from those of IrF<sub>4</sub> and its isostructural neighbors PdF<sub>4</sub> and PtF<sub>4</sub>. The principal distinguishing features of the RhF<sub>4</sub> XRDP were the broadness of the diffraction lines (which were broad even in high temperature preparations, which should have favored recrystallization) and the absence of some lines characteristic of the PdF<sub>4</sub> type pattern [8,9]. The OsF<sub>4</sub> pattern exhibited the same peculiarities. It was now clear that whatever structural changes set RhF<sub>4</sub> apart from PdF<sub>4</sub>, the same also applied to OsF<sub>4</sub>.

# 2.1. A possible new structure type for OsF<sub>4</sub> and RhF<sub>4</sub>

The PdF<sub>4</sub> type structure is like that of rutile with every other metal atom missing in the stacks of edge-shared MF<sub>6</sub> or F<sub>6</sub> octahedra. Fig. 2 represents the (polar) structure as viewed down the stacks of superimposed edge-shared octahedra. Since X-ray scattering, at a particular diffraction angle, is approximately proportional to the atomic number of the atom, the scattering by Os (atomic number = 76) and Rh (atomic number = 45) outweighs that from four F atoms out of phase with the metal atoms  $(4 \times \text{atomic number } 9)$ . We therefore expect that in such a structure 2 2 0 reflections should be observed, and perhaps be even a strong-line in the OsF<sub>4</sub> case. In the earlier work ([9], Table 3), it was noted that this reflection was very weak (vvw) in RhF4, whereas in PdF<sub>4</sub> the same reflection ([9], Table 1), was strong (s). (In XRDPs of RhF<sub>4</sub>, also, the 220 reflection is close to the position of a strong reflection from R-RhF<sub>3</sub> [1], to which impurity the weak observed line in RhF4 could be attributed.) But Rh scattering is not as dominant in RhF<sub>4</sub> as that of Os in OsF<sub>4</sub> so the failure to observe a 220 reflection in the XRDP of the latter provided the decisive indicator that the PdF<sub>4</sub> type structure was not appropriate for these tetrafluorides.

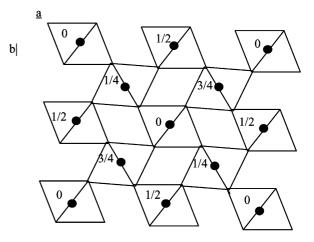


Fig. 2. PdF<sub>4</sub> type structure—projection in the 0 0 1 plane.

Table 1 Pyrochlore RhF<sub>3</sub> X-ray powder data (Cu K $\alpha$ , Ni filter) with a face-centered cubic (probable space group Fd3m) unit cell,  $a_0=10.292(5)\,\text{Å}$ ;  $V=1090\,\text{Å}^3$ 

 $1/d^2 \times 10^4$  $I/I_0$ h k lObserved Calculated 3 1 1 5 1 1; 3 3 3 711;551 731;553 820;644 822;660 751;555 911;753 921;761;655 9 3 5; 7 7 1; 755 1022;666 1111;766 1131;971;955 

Table 2 X-ray powder data (Cu K $\alpha$  radiation, Ni filter) for OsF<sub>4</sub>, F orthorhombic cell with a=9.89(1) Å, b=9.36(1) Å, c=5.70(1) Å, z=8, V=528 Å<sup>3</sup>

<i>I</i> / <i>I</i> <sub>0</sub>	$1/d^2 \times 10^4$		h k l
	Observed	Calculated	
w	431	409	200
		456	020
vs	523	524	111
		865	22(
W	1351	1341	3 1 1
w	1454	1437	1 3 1
S	1636	1635	400
		1639	202
m	1827	1826	040
m	2099	2092	420
		2095	2 2 2
w	2969	2983	113
vw	3269	3262	15
vs	3473	3461	440
		3464	2 4 2
vw	3796	3801	3 1 3
vw	3877	3896	133
W	4699	4690	442
		4714	3 3 3
ms	4940	4919	0 0
		5722	153
		5743	460
vw	5758	5746	262
		5785	2 2 4
m	6579	6539	353
		6555	404
m	6748	6745	0 4 4
w	7330	7304	080
m	8436	8390	444
m	8977	9027	064
W	11544	11525	020
m	12268	12223	084
m	13362	13379	175
VW	13900	13855	484

As has already been noted, the RhF $_4$  and OsF $_4$  XRDPs are strikingly similar, and not only in the absence (or weakness) of the 2 2 0 reflections, but also in the broadness of the general  $h\,k\,l$  reflection lines. The two fluorides evidently share a common structure. Aside from these features that distinguish the XRDPs of these two fluorides from those of

the  $PdF_4$  type, however, the patterns otherwise exhibit a close relationship. All of the patterns can be indexed on the basis of a face-centered orthorhombic cell (see [9] and Table 2). In the  $PdF_4$  type structure (see Fig. 3a) each octahedrally coordinated M (i.e.  $[MF_6]$ ) is edge-shared with an octahedral set of  $[F_6]$  above and below it in the stack.

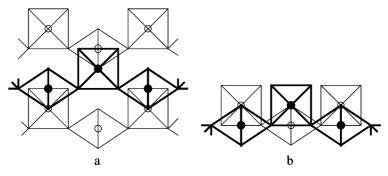


Fig. 3. (a) Polar PdF<sub>4</sub>: 3D structure; and (b) 2D "raft" structure for OsF<sub>4</sub> and RhF<sub>4</sub>.

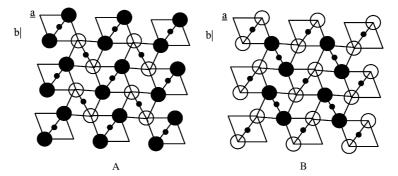


Fig. 4. The two surfaces of the proposed 'raft' structure for OsF<sub>4</sub> and RhF<sub>4</sub>: filled circles for non-bridging F on the top surface, open circles for the same on the bottom surface. Reflecting A in the *ab* plane converts it to B.

The other two F of the  $[MF_6]$  (trans to each other) are shared with two  $[MF_6]$  octahedra. These are oriented roughly at right angles with the first,  $[MF_6]$  and pitched always on the same side (above or below on the  $[MF_6]$  stacking axis, z) to give the three-dimensional polar array represented in Figs. 2 and 3a. This arrangement always has the cis non-bridging F of each  $[MF_6]$  on the same side (up, or down, on the stacking axis, z). It will be recognized, from Fig. 3b, that if linkage of the  $[MF_6]$  is changed, so that the cis non-bridging F ligands on adjacent  $[MF_6]$  are alternately up and down, a two-dimensional ('raft') structure results.

Close-packing of such 'rafts' requires that the non-bridging F of one such 'raft' inter-lock with the adjacent nonbridging F of the adjacent 'raft'. Fig. 4 represents these non-bridging F on the surface of each raft, filled circles for those on one side and open circles for those on the other. For each face-to-face contact there are two stacking possibilities. These can be visualized with the two 'unit cells' of the raft structure labeled A and B. (Reflecting A in the ab plane produces arrangement B.) Let the filled circles represent up F and the open down F. The up of A is the down of B, and vice versa. If we stack A on A or B on B (filled circles over filled, unfilled over unfilled), we see that the M atoms would be superimposed. To mesh A with B it is necessary to translate A or B by  $\pm 1/4a$ . If we mesh A with B or vice versa, we therefore see that the M do not superimpose. For the latter arrangements the projection of the M atom array into the ab plane is represented by a simple rectangle of sides quarter a, quarter b.

The consequence of randomly stacking 'rafts' A and B therefore satisfies our need to have no significant diffraction from the 2 2 0 planes. Also, since the fit A on A (or B on B) is not exactly the same as A with B, the inter-raft separation will not be exactly the same for both. That accounts, at least partially, for the line broadening in the XRDP.

# 2.2. Why the structural variety?

It is clear from the non-equivalence of the RuF<sub>4</sub> and OsF<sub>4</sub> structures on the one hand, and the RhF<sub>4</sub> and IrF<sub>4</sub> on the other, that the number of electrons in the d electron configuration has no important influence on which structure is adopted.

Certainly, the effective nuclear charge increases from left to right in each transition series. Also, that increase with atomic number is more dramatic for the second series than for the first, as the molecular instability and oxidizing patterns of the hexafluorides indicate [22]. We therefore expect the F ligands to be more highly polarized in the second series, by the nuclear charge of M, than in the third. This should render the F ligands of the second series fluorides smaller than those of their third series relatives. Certainly PdF<sub>4</sub> [8,9] has a smaller formula-unit volume than isostructural PtF<sub>4</sub> [6,7], 63.20 versus 63.57 Å<sup>3</sup>, which is almost the same as the difference in the formula-unit volumes of the metals themselves (14.71 versus 15.09  $\mathring{A}^3$ ) [23]. The degree of covalency in RhF<sub>4</sub> is therefore likely to be greater than in IrF<sub>4</sub>, and greater in RuF<sub>4</sub> than in OsF<sub>4</sub>. But, if so, the degree of covalency in the M-F bonding does not determine the structure type, since PdF<sub>4</sub> should have higher M-F covalency than RhF4, which should be more covalent than IrF<sub>4</sub>, which has the same structure as

Table 3 X-ray powder data (Cu K $\alpha$  radiation, Ni filter) for ReF<sub>4</sub> (F orthorhombic, pseudo-tetragonal, with a=b=9.61(2) Å, c=5.66(1) Å, z=8, V=522 Å<sup>3</sup>, possible space group Fdd2)

$I/I_0$	$1/d^2\times 10^4$		h k l
	Observed	Calculated	
s	537	529	111
w(br)	867	866	220
m(br)	1401	1395	3 1 1; 1 3 1
s	1696	1683	202;022
_	_	1732	400;040
vw	2138	2116	222
_	_	2165	420; 240
vw	2240	2261	3 3 1
w(br)	3048	3029	113
_ ` ´	_	3127	511;151
vs(br)	3450	3414	422;242
. ,		3465	440
w(br)	3920	3892	3 1 3; 1 3 3
_ ` ´	_	3994	531;351
_	_	4331	620; 260
vvw	4723	4758	3 3 3
w(br)	4999	4994	004

PdF<sub>4</sub>! Moreover, ReF<sub>4</sub>, on the left in the third transition series appears to have the PdF<sub>4</sub> structure type (see Fig. 2, and Table 3). If this is established, there will be no clear trend for the structure type dependence on Periodic Table location. Here we have a puzzle for future workers to sort out!

#### 3. Experimental

Apparatus and methods for the work using liquid HF were as described in [21] and references therein.

For the aHF-based work the reaction vessels routinely used were of three types. Single armed reaction vessels consisting of a FEP tube sealed at one end and joined to a Teflon valve and two armed reactors consisting of two such FEP tubes, joined to a Teflon valve by means of a Swagelok Teflon tee piece. Such apparatus was used for reactions involving aHF in which pressures would not exceed 2 atm FEP tubes were sealed by heating one end over a flame, pinching it shut with pliers when it had softened as signaled by increased transparency. Teflon valves, Fig. 5, consisted of a Teflon valve body, a Kel-F stem containing an inserted Teflon stem tip, and Teflon packing washers. These valves were always attached to the apparatus with the seat side of the valve exposed to vapors from highly acidic oxidizing solutions, which could only contact the Teflon tip of the stem, when the valve was closed. The Teflon packing nuts were pressed against the stem and valve body, to make a seal, by means of an aluminum backing washer and a screw tightened nut. Connections were made to the inlet and outlet of the valve by means of 1/4" Swagelok fittings, or through 1/4" FEP tubing sweated on to a knurled inlet tube at the seat side of the valve.

# 3.1. Drybox

Involatile, air-sensitive materials were handled in the dry Ar atmosphere of a Vacuum Atmospheres Corp., DRILAB. Freedom of the atmosphere from significant oxygen, or water contamination was judged by the continued burning of an incandescent filament.

# 3.2. Reagents

Anhydrous hydrogen fluoride (aHF) 98% or better, as obtained from Matheson (Newark, CA) was dried using either dioxygenyl salts [24] ( $O_2SbF_6$ , or  $O_2Sb_2F_{11}$ ), or  $K_2NiF_6$  (Ozark–Mahoning–Pennwalt).

The commonly used reagent gases  $BF_3$  (Matheson),  $PF_5$ , and  $AsF_5$  (Ozark–Mahoning–Pennwalt, Tulsa, OK) were each checked before use by IR spectroscopy and when found to be free of major volatile impurities were used without further purification. The much less volatile liquids  $BrF_3$  (Matheson) and  $SbF_5$  (Ozark–Mahoning) were generally distilled out of their cylinders directly into reaction vessels. Bromine usually accompanied  $BrF_3$  distillation, however, since this served as a useful diluent and was a by-product of  $BrF_3$  oxidations, further purification was not carried out. When small amounts of volatile  $O_2SbF_6$  impurity in the  $SbF_5$  could not be tolerated, it was removed by exposing the distilled  $SbF_5$  to  $C_6F_6$  (PCR, Gainesville, FL) which was oxidized to  $C_6F_6^+$  by any  $O_2^+$  present [25].

# 3.3. X-ray powder diffraction photograph (XRDP)

Debye–Scherrer photographs were taken on General Electric Precision Powder cameras (45 cm circumference) using Ni filtered, Cu  $K\alpha$  radiation. Powder samples for

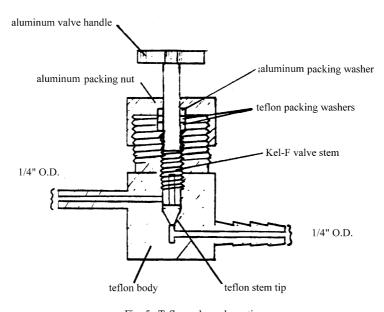


Fig. 5. Teflon valve schematic.

XRDP were loaded into 0.3 or 0.5 mm diameter quartz capillaries (Charles Supper Co., Natick, MA) inside the DRILAB and temporarily sealed with Kel–F grease (Halocarbon Product Corp., N. Augusta, SC). The capillaries were finally sealed outside the DRILAB by drawing down using a micro-torch.

### 3.4. Magnetic measurements

Variable temperature magnetic susceptibility data were collected on a SQUID magnetometer (S.H.E. Corp., and Quantum Dynamics). Samples were held in cylindrical containers made of two Kel–F caps, one fitting inside the other. These were passivated with 2 atm of a F<sub>2</sub>/AsF<sub>5</sub>, or F<sub>2</sub>/BF<sub>3</sub> mixture overnight before use. Samples were made up in the DRILAB. Typically measurements were taken at 5 and 40 kG between 6 and 280 K.

#### 3.5. Specific syntheses

#### 3.5.1. Pyrochlore-form RhF<sub>3</sub>

The preparation of the BrF<sub>3</sub> adduct RhF<sub>3</sub> (which was shown to be diamagnetic in this work) and its decomposition to purple–red pyrochlore-form RhF<sub>3</sub> followed the method described by Sharpe [18] for 'RhF<sub>4</sub>'. Table 1 gives the indexing of the XRDP for this material.

#### 3.5.2. Preparation of $OsF_4$

 $K_2OsF_6$  [13] (600 mg, 1.5 mmol) in a T assembly was dissolved in aHF (6 ml) at room temperature. Addition of AsF<sub>5</sub> gave an immediate precipitate of a dense yellowbrown solid. When the solution was colorless, AsF<sub>5</sub> addition was stopped. (KAsF<sub>6</sub> side product has low solubility in the presence of dissolved AsF<sub>5</sub>.) Decantation and back distillation [21] was repeated 10 times to remove KAsF<sub>6</sub> from the OsF<sub>4</sub>. The aHF and any AsF<sub>5</sub> were evacuated through a sodalime scrubber dedicated to arsenic waste. A nearly quantitative yield of OsF<sub>4</sub> was obtained, there being some loss in the decantation steps. X-ray powder data, Table 2, were indexed completely on the basis of an F orthorhombic unit cell. Variable temperature magnetic susceptibility measurements indicated a low, approximately temperature independent paramagnetism, consistent with known octahedral, third transition series d<sup>4</sup> systems.

# 3.5.3. $K_2OsF_6$ with $BF_3$

 $BF_3$  added to a solution of  $K_2OsF_6$  (492 mg, 1.3 mmol, in 5 ml aHF) gave an orange insoluble solid (207 mg). This was amorphous to X-rays. Addition of  $AsF_5$  in aHF converted this to yellow–brown  $OsF_4$ , as indicated by XRDP.

# 3.5.4. $K_2RuF_6$ with $BF_3$

Interaction of  $K_2RuF_6$  (600 mg, 2.0 mmol) with  $BF_3$  gave an insoluble orange solid with no XRDP. Interaction of this with  $AsF_5$  in aHF produced deep pink  $RuF_4$ , proved by XRDP [3].

# 3.5.5. $K_2ReF_6$ with $AsF_5$

Addition of  $AsF_5$  to a pale pink solution of  $K_2ReF_6$  [13] in aHF precipitated a pale pink solid. It quickly dissolved ( $\sim$ 5 min) in  $AsF_5$ -rich aHF to a green solution, indicating that oxidation of  $ReF_4$  to green  $ReF_5$  by  $AsF_5$  had occurred [26].

# 3.5.6. Preparation of ReF<sub>4</sub>

Excess of SbF<sub>5</sub> (1 ml, 14 mmol) in aHF was poured into an aHF solution of  $K_2ReF_6$  (295 mg, 0.78 mmol) to give a pink precipitate. In  $\sim$ 30 min some of the pink solid dissolved in the SbF<sub>5</sub>-rich solution and gave a pale blue solution. The remaining solid was washed in the usual way to leave 110 mg (0.32 mmol, 42% yield based on  $ReF_4$ ). The XRDP data (Table 3) were indexed on the basis of a of  $PdF_4$  type cell.

#### 3.5.7. Preparation of MoF<sub>4</sub>

MoF<sub>4</sub> was made in the same manner as ReF<sub>4</sub>. To a green solution of  $K_2MoF_6$  [13] (410 mg 1.4 mmol) in aHF, in a T reactor, a solution of SbF<sub>5</sub> in aHF was slowly decanted to produce a pale green precipitate. Addition of the SbF<sub>5</sub>/aHF solution, beyond the point at which the supernatent became colorless, caused some dissolution of the green solid, producing a pale green solution. This product was washed and dried in the usual way [21]. The green solid (110 mg, 0.58 mmol, 41% yield based on MoF<sub>4</sub>) proved to be very poorly crystalline, but the X-ray powder data matched the strongest lines in the pattern described by Payne and Asprey [11] for their MoF<sub>4</sub>. It was not indexed.

# 3.5.8. $K_2MoF_6$ with $AsF_5$

Addition of AsF<sub>5</sub> to a solution of K<sub>2</sub>MoF<sub>6</sub> in aHF produced a green solid which on standing in AsF<sub>5</sub>/aHF solution quickly became pale, tan, indicative of MoF<sub>5</sub> formation [27].

# 3.5.9. Preparation of PdF<sub>4</sub>

 $K_2PdF_6$  (prepared by the method of Sharpe [19]) gave a yellow–orange solution in aHF. Addition of excess  $AsF_5$ , with rapid stirring of the solution, quickly produced a brown solid. With the reactor continuously agitated and a high  $AsF_5$  concentration maintained ( $\sim$ 12 h) in the AHF, the brown solid gradually took on the pink to red color typical of  $PdF_4$  [9]. XRDP of the pale red solid showed it to be identical to those of  $PdF_4$  prepared by high-pressure fluorination.

# Acknowledgements

The authors gratefully acknowledge the support of the bulk of this work by the Director, Office of Basic Energy Sciences, and Chemical Sciences Division of the US Department of Energy under contract number DE-AC-03-76SF00098. The pyrochlore-form RhF<sub>3</sub> (previously believed to be RhF<sub>4</sub>) was prepared at U.B.C. with support from the National Research Council of Canada.

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